

Low-Resistivity Indium-Tin-Oxide Transparent Conducting Films: Dependence of Carrier Electron Concentration on Tin Concentration

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Abstract. Indium-Tin-Oxide (ITO, tin-doped In₂O₃) films with low resistivity (7.7×10^{-5} ohm cm) and high carrier electron concentration (1.8×10^{21} cm⁻³) was successfully prepared by spray chemical vapor deposition in air and post-deposition annealing in reducing atmosphere in our previous papers; Y. Sawada et al., Thin Solid Films, 409 (2002) 46-50 and Y. Sawada, Materials Sci. Forum, 437-438 (2003) 23-26. Doping one tin ion generated two carrier electrons at low concentration of tin. The relation between carrier electron concentration and tin concentration are discussed in the present paper to propose a noble defect model.

Introduction

Transparent conducting films of tin-doped In₂O₃ (Indium-Tin-Oxide, ITO) are used for solar cells, flat panel displays including touch panels etc. and usually deposited by magnetron sputtering. The authors reported elsewhere [1, 2] low-resistivity ITO films fabricated by spray chemical vapor deposition very cheaply. The lowest resistivity of the as-deposited in air was 1.8×10^{-4} Ω cm. This value was compatible with those deposited by magnetron sputtering. The post-deposition annealing in reducing atmosphere lowered the resistivity. The lowest resistivity was 7.7×10^{-5} Ω cm. In the present paper the carrier electron concentration of ITO films after the post-deposition annealing will be discussed as a function of tin concentration.

Experimental

The preparation and evaluation of the films were reported elsewhere [1, 2] in detail. Ethanol solution of InCl₃·2.7H₂O and SnCl₂·0.86H₂O was sprayed onto a glass substrate (Corning 7059) heated at 300-350°C on a hotplate to deposit ITO films (approx. 200 nm). The films were annealed at 600°C for 2 h in N₂-0.2% H₂ gas flow (300 mL/min) to lower the resistivity.

Results and Discussion

Figure 1 shows carrier electron concentration (n) of indium oxide films as a function of tin concentration (C_{Sn}) after annealing in reducing atmosphere. The C_{Sn} is tin concentration of

$100 \times [\text{Sn}] / ([\text{Sn}] + [\text{In}])$, where $[\text{Sn}]$ and $[\text{In}]$ are the concentration (cm^{-3}) of tin and indium atoms, respectively, in indium oxide crystal.

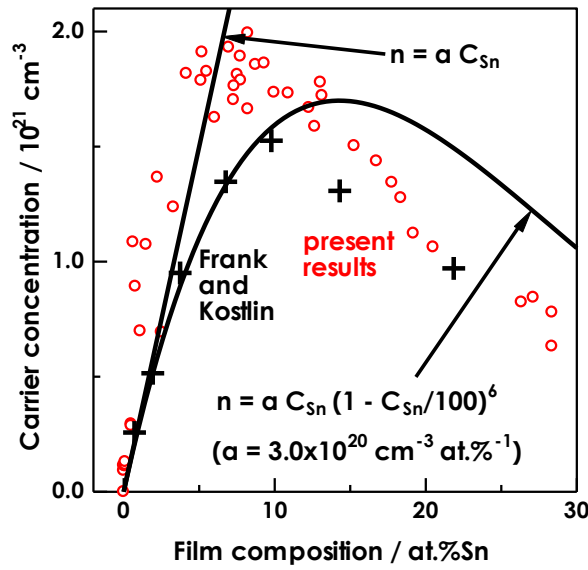


Fig. 1. Carrier electron concentration of indium oxide films as a function of tin concentration after annealing in reducing atmosphere: comparison with the results by Frank and K östlin

This figure also shows the results by Frank and K östlin [3] who deposited by spray CVD and annealed in reducing atmosphere. They interpreted that a Sn^{4+} ion at a In^{3+} site generates one carrier electron when interaction is negligible between the two neighbouring Sn^{4+} ions at low concentration of tin. This is expressed as Eq. 1 by Kr öger-Vink notation [4] at low tin concentration.



Carrier electron concentration (n) is expressed as

$$n = [\text{Sn}] = a C_{\text{Sn}} \quad (2)$$

The constant $a = 3.0 \times 10^{20} \text{ cm}^{-3} \text{ at.\%}^{-1}$ is determined from the lattice constant ($1.0118 \times 10^{-7} \text{ cm}$) of In_2O_3 and number of metal ions (32) in a unit cell. At higher concentration of tin, carrier electron concentration is deviates from Eq. 2. They attributed to a formation of neutral defect complex $\text{Sn}_{\text{In}}^{\bullet} 2 \text{O}_{\text{i}}^{\times}$. They assumed two nearest-neighbor Sn^{4+} ions at In^{3+} site coupled with one interstitial O^{2-} ions to eliminate carrier electrons as Eq. 3.



The $\text{O}_{\text{i}}^{\times}$ is an interstitial oxygen ion (O^{2-}) located at a space called quasi-anion site of In_2O_3 lattice (C-rare earth oxide structure). They assumed that tin ions occupy only at “b-site” of In_2O_3 lattice and proposed Eq. 4 with 6 nearest-neighbor tin ions.

$$n = a C_{\text{Sn}} (1 - C_{\text{Sn}}/100)^6 \quad (4)$$

This equation satisfies their results at \leq approx. 10 at.%Sn but fails to explain higher carrier electron concentrations of the present results.

The present results are shown in Figure 2 in order to compare with the results by Ohta et al. who deposited ITO film on a single crystal of zirconia by expensive pulsed laser deposition [5]. The

present highest value ($1.8 \times 10^{21} \text{ cm}^{-3}$) agreed with their ones and our previous result [1, 2] at C_{Sn} , 5-7 at.%. The results fit with Eq. 5 at low concentration of tin (< approx. 3 at.%).

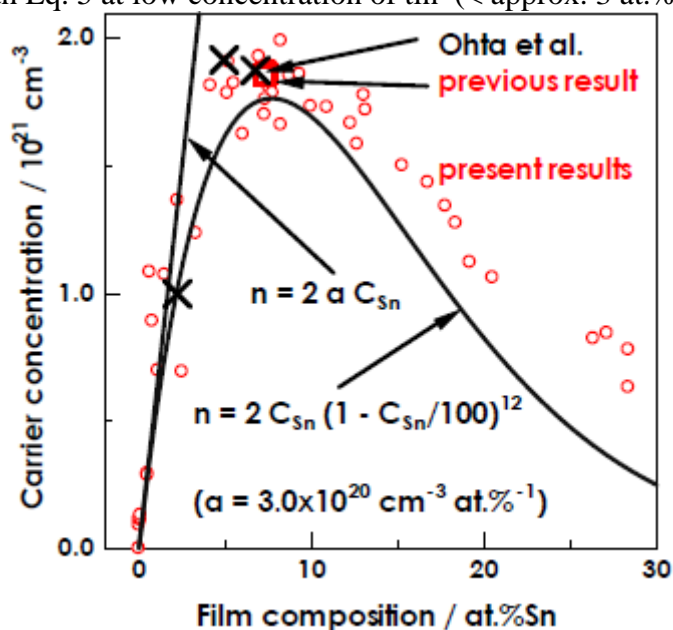


Fig. 2. Carrier electron concentration of indium oxide films as a function of tin concentration after annealing in reducing atmosphere: comparison with the results by Ohta et al. and our previous result.

$$n = 2 [\text{Sn}] = 2 a C_{\text{Sn}} \quad (5)$$

This assumes that doping one tetravalent tin ion (Sn^{4+}) generates two carrier electrons (e^-) although it seems rather strange. In order to explain the present high carrier concentration, simultaneous formation of Sn_{In} and V_{O} was tentatively assumed to generate two carrier electrons as Eq. 6



V_{O} is an oxygen ion vacancy coupled with an electron. A novel model with 12 nearest-neighbor tin ions (both b- and d-site) is proposed.

$$n = 2 a C_{\text{Sn}} (1 - C_{\text{Sn}}/100)^{12} \quad (7)$$

This is a better fit with the present results although the defect mechanism explaining the phenomenon is not clear at present.

Structure and other properties of the present films will be demonstrated: the resistivity, carrier electron mobility, transmittance and reflectance, X-ray diffraction, FE-SEM photos etc.

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